### Predicting the Service-Life of Concrete Structures Exposed to Chemically Aggressive Environments

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Materials Service Life Laval University

CEMENTITIOUS MATERIALS FOR WASTE TREATMENT, DISPOSAL, REMEDIATION & DECOMMISSIONING WORKSHOP

Savannah River Site, Aiken South Carolina December 12-14, 2006

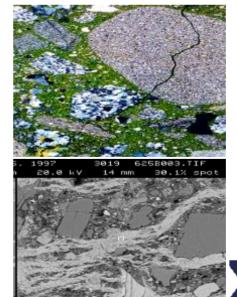


# **Concrete degradation**













### **SUMMA Consortium - Members**























The transport of ions in **STADIUM®** is modeled with the extended Nernst-Planck equation with an advection term:



STADIUM®

**Mass conservation equation:** 

$$\frac{\partial(wc_i)}{\partial t} + \operatorname{div}(j_i) = 0$$

#### Variables:

- Concentrations c<sub>i</sub>
- Diffusion potential  $\psi$
- Water content w
- Temperature *T*



To complete the system of equations, the following relationships are considered:

Poisson: 
$$\tau \frac{d}{dx} \left( w \frac{d\psi}{dx} \right) + \frac{F}{\epsilon} w \left( \sum_{i=1}^{N} z_i c_i \right) = 0$$

Richards: 
$$\frac{\partial w}{\partial t} - \frac{\partial}{\partial x} \left( D_w \frac{\partial w}{\partial x} \right) = 0$$

Heat conduction: 
$$\rho C \frac{\partial T}{\partial t} - div \ (k \text{grad} T) = 0$$

The system of equations is solved using the finite element method:

- 11 unknowns:  $8 \times c_i + w + \psi + T$
- 11 equations: 8 conservation + Poisson + Richards + Heat



The chemical reactions are modeled according to dissolution/precipitation equilibrium relationships:

Dissolution/precipitation: 
$$K_m = \prod_{i=1}^N c_i^{\nu_{mi}} \gamma_j^{\nu_{mi}}$$

Portlandite: 
$$K_{\rm CH} = \gamma_{\rm Ca} \gamma_{\rm OH}^2 [{\rm Ca}] [{\rm OH}]^2$$

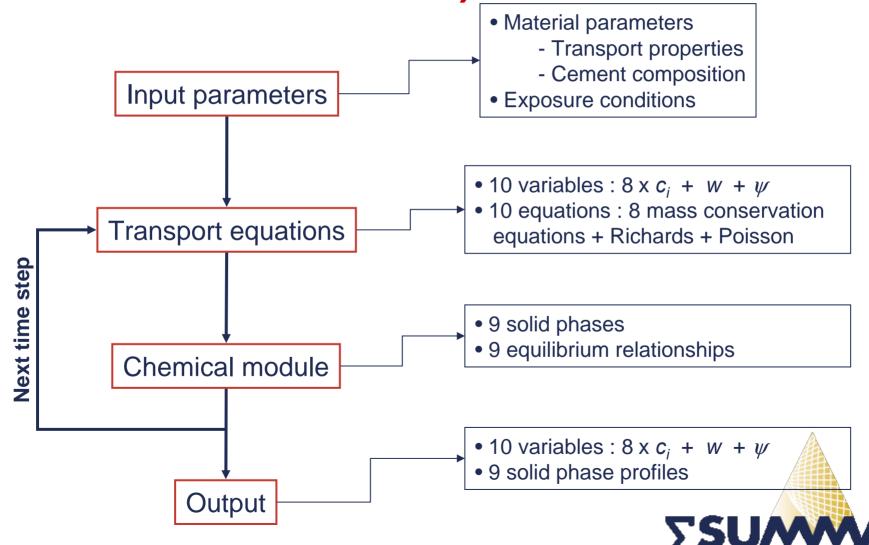
In the case of chlorides, the formation of Friedel's salts is modeled according to an ionic exchange relationship:

• Friedel's salts formation: 
$$\underbrace{(AFm^+\text{-}SO_4^{2\text{-}}) + 2Cl^- \rightarrow (AFm^+\text{-}2Cl^-) + SO_4^{2\text{-}}}_{Monosulfates} + \underbrace{(AFm^+\text{-}2Cl^-) + SO_4^{2\text{-}}}_{Friedel's \ salt}$$

• Relationship: 
$$K = \frac{\{\text{Cl}\}^2}{\{\text{SO4}\}} \frac{\left[\text{AFm}_{SO4}\right]}{\left[\text{AFm}_{Cl}\right]}$$



Operator-splitting algorithm (Isothermal case - SNIA)



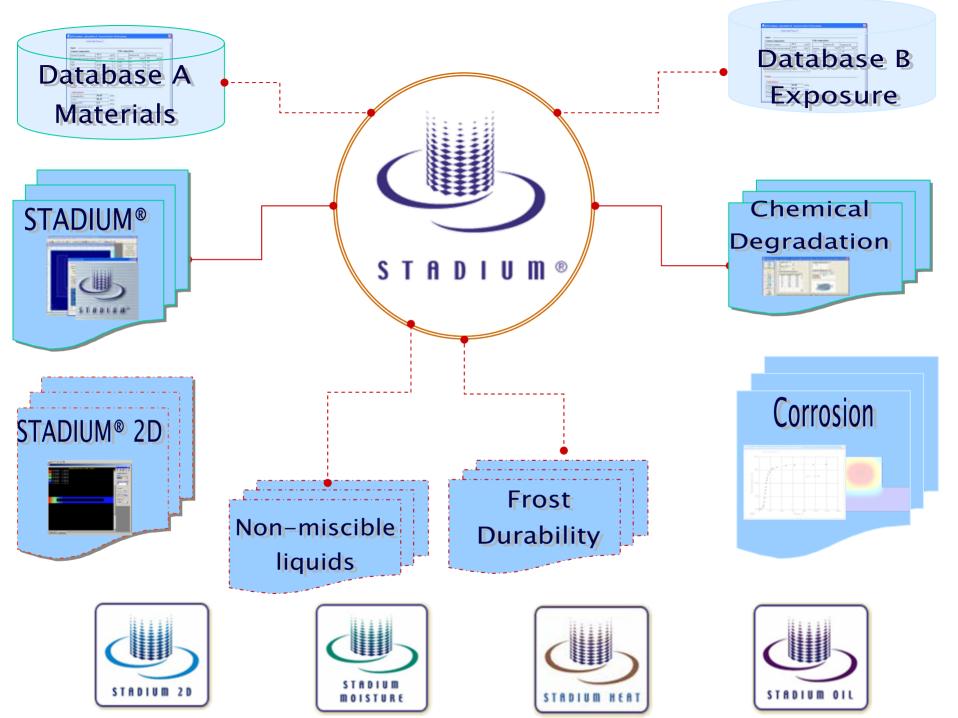
To bring the solution back to equilibrium at one node, the following non-linear system of equations is solved:

$$K_{CH} = \gamma_{Ca} \gamma_{OH}^2 (Ca^\circ + X_{CH} + 6X_{Aft} + X_{Gyp} + ...) (OH^\circ + 2X_{CH} + 4X_{Aft} + ...)^2$$
 $K_{Gyp} = \gamma_{Ca} \gamma_{SO4} (Ca^\circ + X_{CH} + 6X_{Aft} + X_{Gyp} + ...) (SO_4^\circ + 3X_{Aft} + X_{Gyp} + ...)$ 
 $\vdots$ 

 $X_i$  = amount of solid i dissolved or formed  $C^\circ$  = concentration before equilibrium

Numerical method: Newton-Raphson





## Input parameters

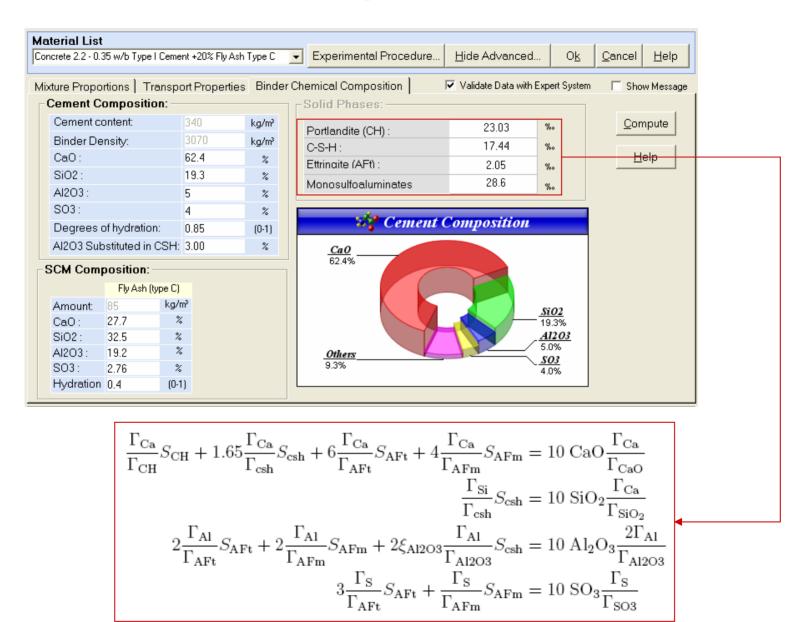
In order to run the model, one needs to generate information on the following properties:

- Mixture characteristics
- Binder composition
- Porosity
- Pore solution
- Diffusion coefficients (formation factor)
- Water diffusivity

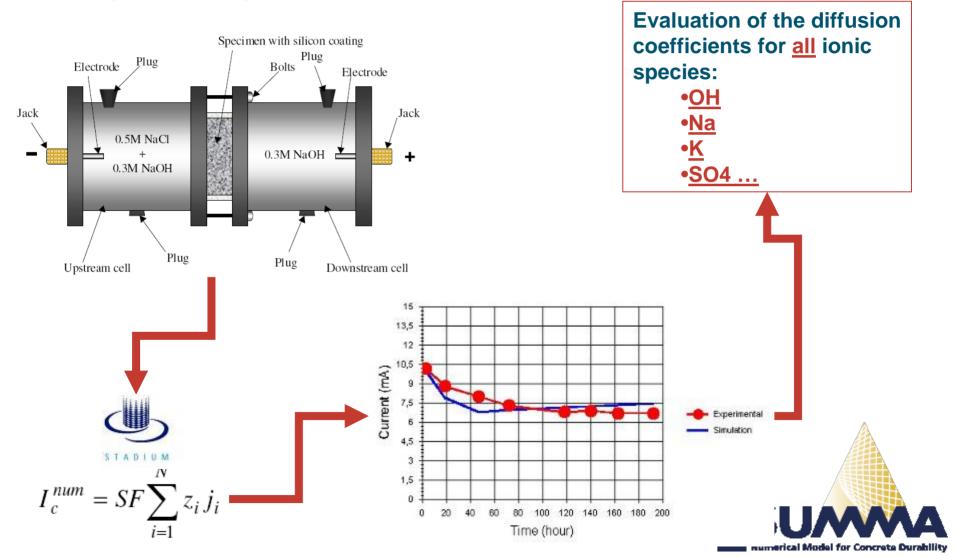
The other parameters are either physical constants or properties that have a relatively weak influence on the output:

- •F, R, z<sub>i</sub>, ...
- •Thermal properties: k, C

# **Cement chemistry**



#### Experimental procedure:



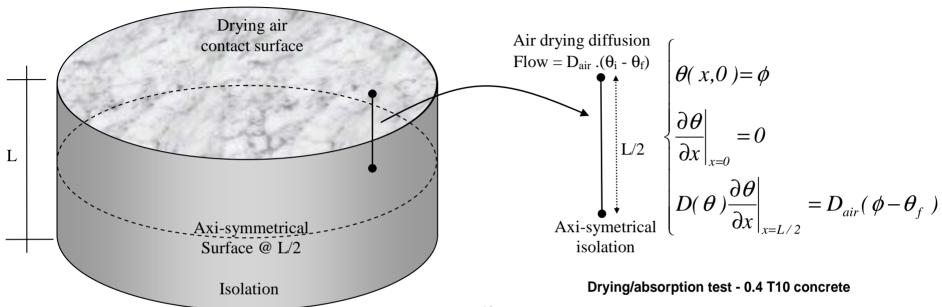
ASTM Type I 
$$w/c = 0.5 - Cure = 18 m.$$

Test Condition		au
Solution	V/mm	
NaCl - 0.5 M	0.4	35.4
NaCl - 0.5 M	0.2	38.5
NaCl - 0.1 M	0.4	39.0
$Na_2SO_4 - 0.2 M$	0.4	41.0

Variation =  $\pm 14\%$ 



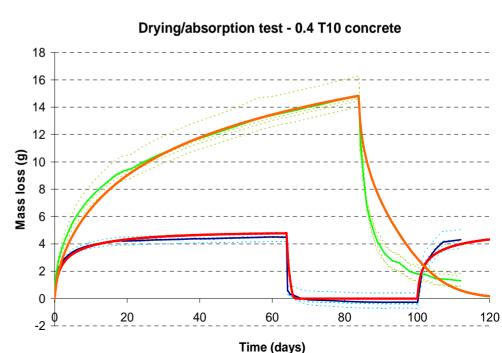
### **Water diffusivity**



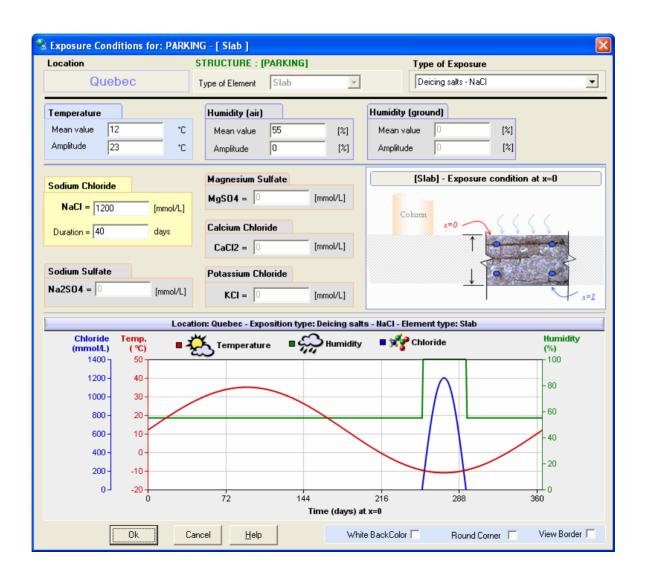
$$D = D_0 \exp(Bw)$$

B = 80 for all

$$D_0 = 1.5 \times 10^{-14} \text{ (w/c} = 0.45)$$
  
 $D_0 = 15 \times 10^{-14} \text{ (w/c} = 0.75)$ 

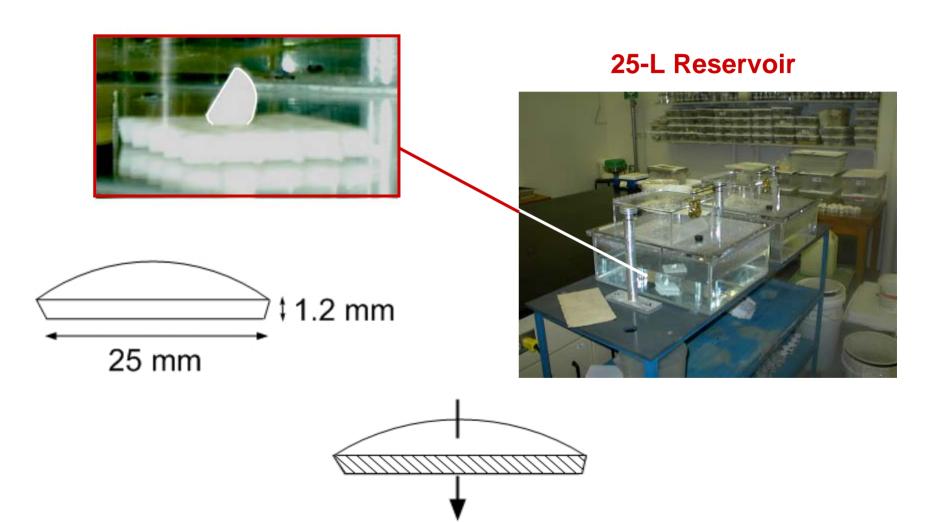


### **Exposure conditions**



# **Experimental validation**

Thin C<sub>3</sub>S slices

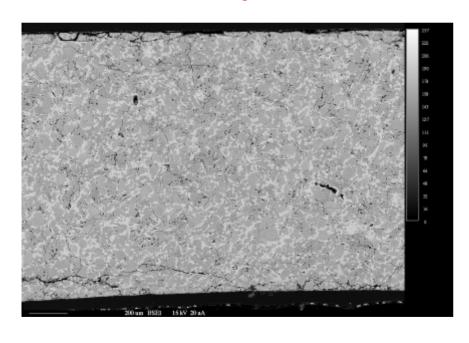


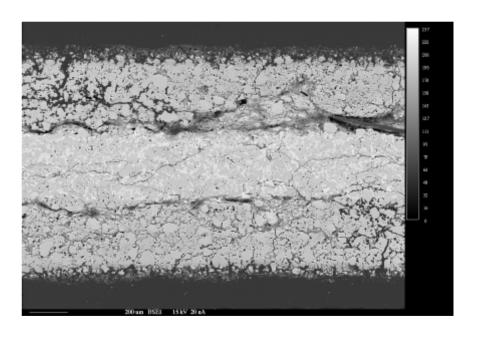
# **Validation - Leaching**

### Degradation analyses

Sound C<sub>3</sub>S paste

Leached C<sub>3</sub>S paste





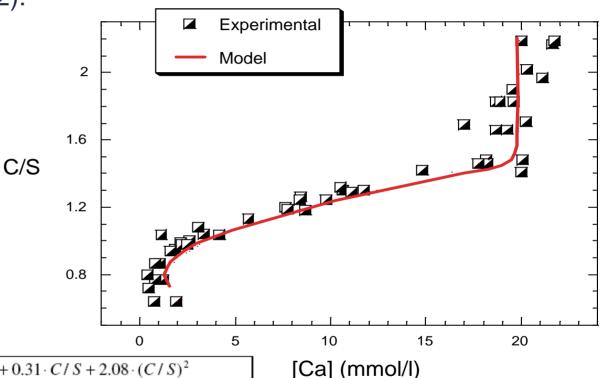
Initial Porosity = 50%  $\tau = 35.2$ 

# **Modeling - Leaching**

The C-S-H decalcification modeling is based on

Berner's approach (1992):

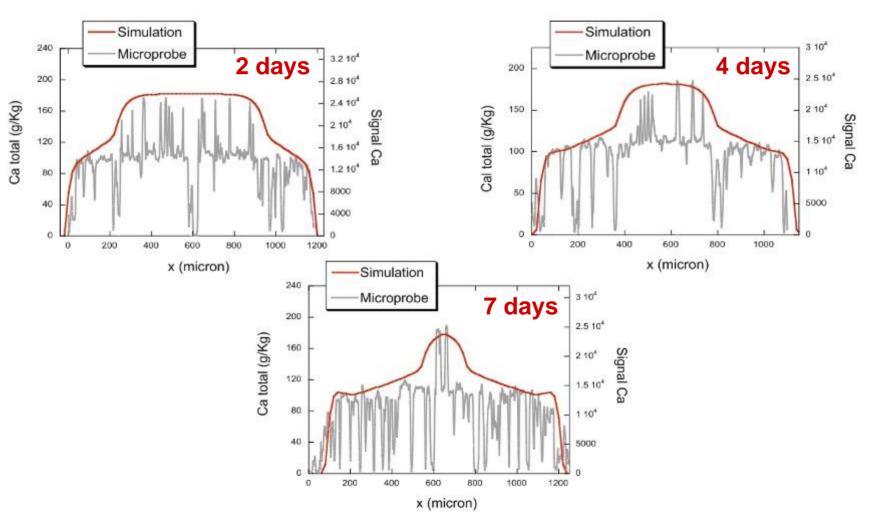
 $Ca^{2+}$   $Ca(OH)^{+}$   $H_4SiO_4$   $H_3SiO_4^{-}$  $H_2SiO_4^{2-}$ 



C/S ≤ 0.7	$pK_{SiO2} = 2.74 + 0.31 \cdot C/S + 2.08 \cdot (C/S)^{2}$ $pKcsh = 7.50$
$0.7 \le C/S \le 1.5$	$pKcsh = 5.29 + 3.01 \cdot C/S - 0.51 \cdot (C/S)^{2}$ $pK_{Ca(OH)2} = 18.54 - 16.58 \cdot C/S + 5.15 \cdot (C/S)^{2}$
C/S≥1.5	$pKcsh = 8.60$ $pK_{Ca(OH)2} = 5.18$

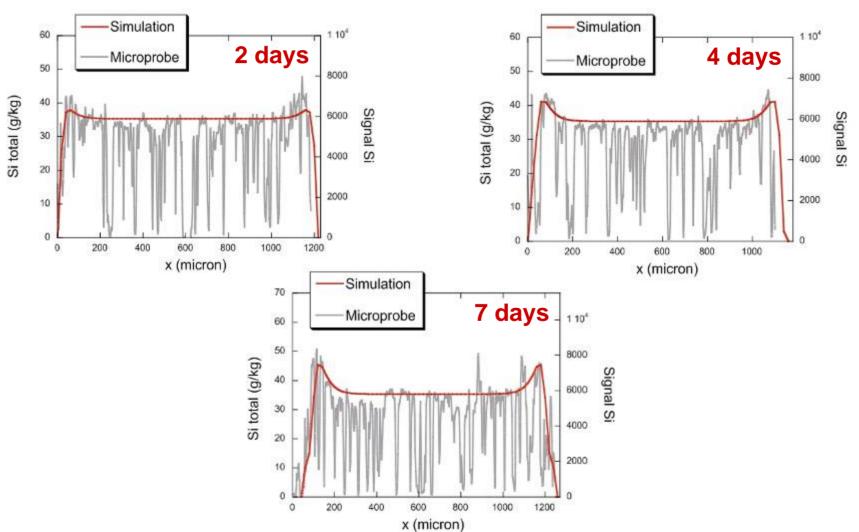
## **Validation - Leaching**

Thin C<sub>3</sub>S slices (w/c: 0.5) – Pure water Calcium profiles



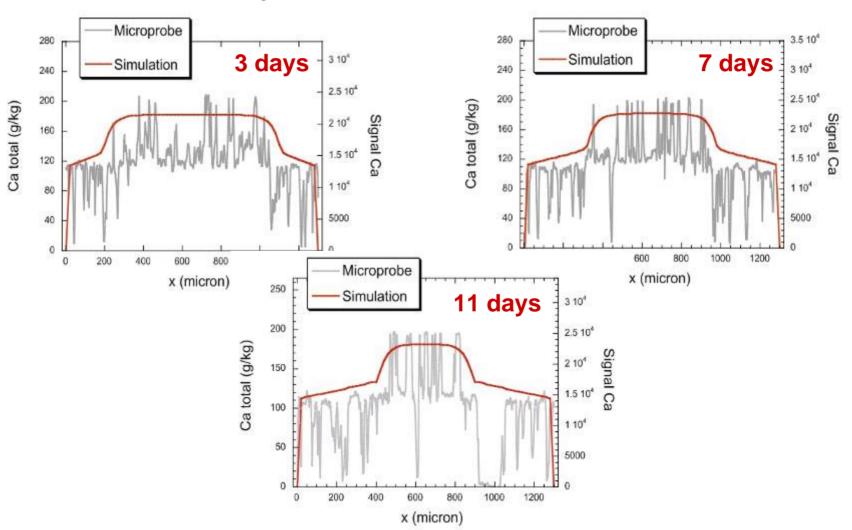
# **Validation - Leaching**

Thin C<sub>3</sub>S slices (w/c: 0.5) – Pure water Silicon profiles



# Validation – Leaching (pH = 12)

Thin C<sub>3</sub>S slices (w/c: 0.5) - Calcium profiles



# Validation - Leaching

Solid phase	Chemical description	Equilibrium relationship	-log(Ksp)
Portlandite	Ca(OH) <sub>2</sub>	$K_{sp} = {Ca} {OH}^2$	5.2
C-S-H	1.65CaO.SiO <sub>2</sub> .(2.45)H <sub>2</sub> O *	$K_{sp} = {Ca} {OH}^{2 **}$	6.2
Ettringite	3CaO.Al <sub>2</sub> O <sub>3</sub> .3CaSO <sub>4</sub> .32H <sub>2</sub> O	$K_{sp} = {Ca}^6 {OH}^4 {SO4}^3 {Al(OH)_4}^2$	44.0
Monosulfates	3CaO.Al <sub>2</sub> O <sub>3</sub> .CaSO <sub>4</sub> .12H <sub>2</sub> O	$K_{sp} = {Ca}^{4} {OH}^{4} {SO4} {Al(OH)_{4}}^{2}$	29.1
Gypsum	CaSO <sub>4</sub> .2H <sub>2</sub> O	$K_{sp} = \{Ca\} \{SO4\}$	4.6
Mirabilite	Na <sub>2</sub> SO <sub>4</sub> .10H <sub>2</sub> O	$K_{sp} = {Na}^2 {SO4}$	1.2

<sup>{...}</sup> indicate chemical activity

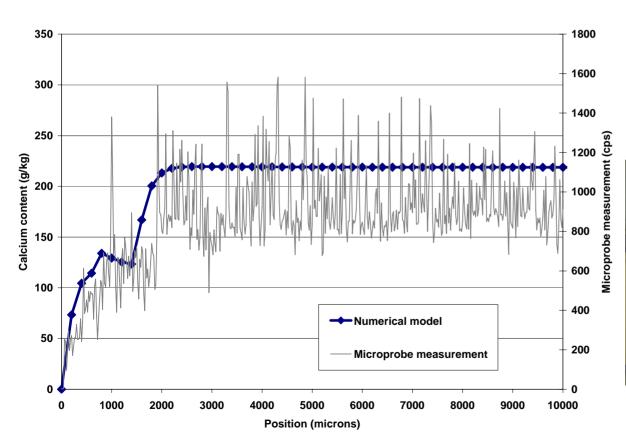
<sup>\*:</sup> A C/S of 1.65 is assumed for the C-S-H

<sup>\*\* :</sup> The C-S-H decalcification is modeled as portlandite dissolution with a lower Ksn

# Validation - Leaching

The model has been validated for several degradation cases.

Pure water exposure:



Paste (w/c:0.6, Type <u>10</u>) exposed to deionized water for 3 months –

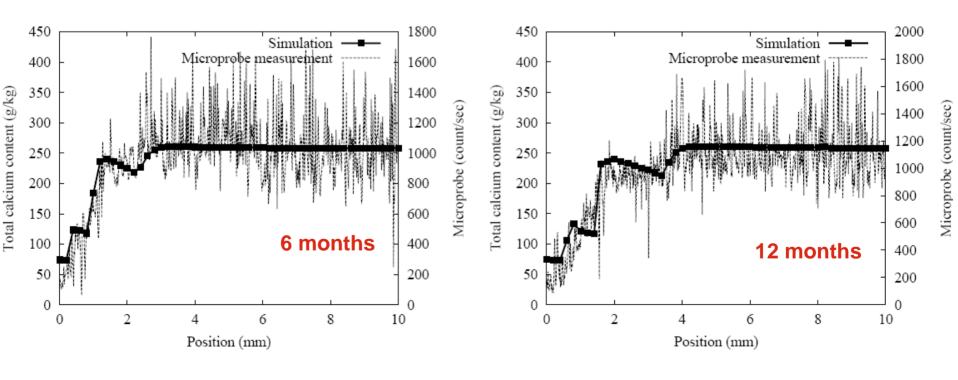
**Calcium profile** 



### Validation – Sulfate attack

Neat Cement Paste (w/c: 0.6, ASTM Type I cement,  $Na_2SO_4 = 50$  mmol/L)

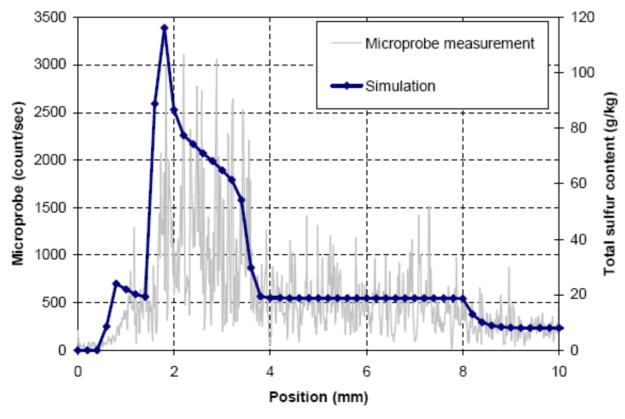
#### **Calcium profiles**



### Validation – Sulfate attack

The model has been validated for several degradation cases.

Sodium sulfate exposure:



Paste (w/c: 0.6, Type 10) exposed to Na<sub>2</sub>SO<sub>4</sub> (50 mmol/L) for 12 months –

Sulfur profile

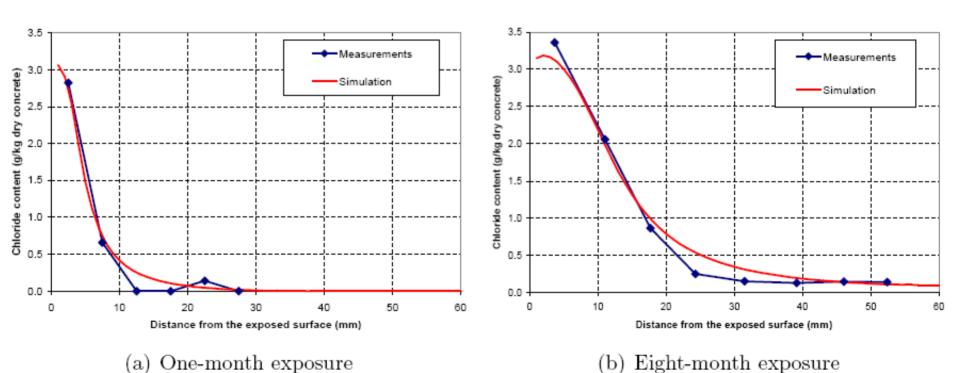


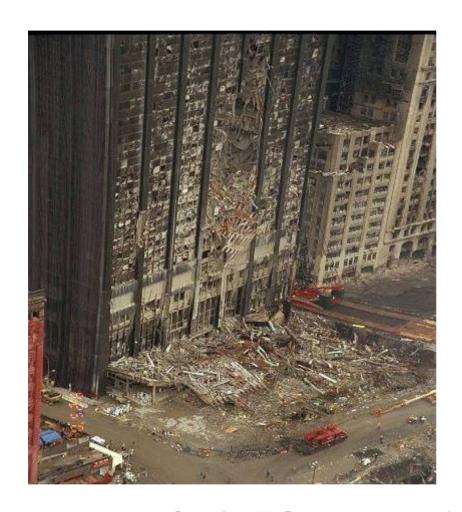
### Validation – Chloride attack

The approach has been validated for several degradation cases.

Sodium chloride exposure:

0.45 CSA Type 10 (ASTM Type I) concrete





130 Liberty St. (WTC New York)

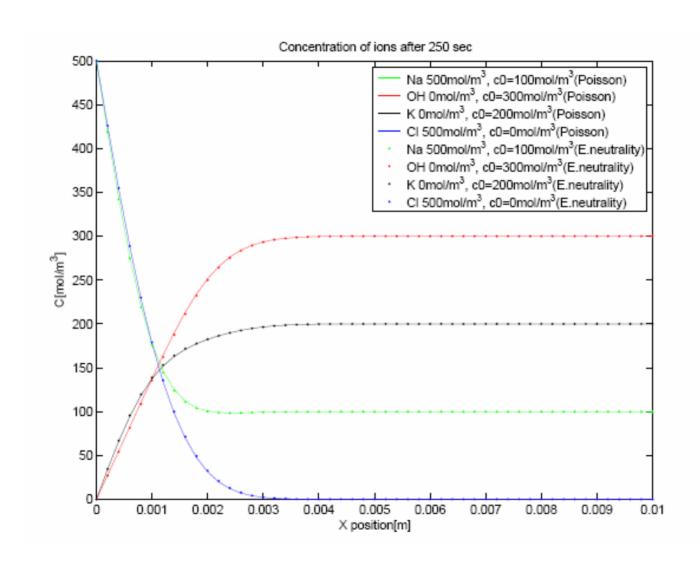


U.S. Embassy (Abu Dhabi)



Parking structure (Louisville, KY)

# **Questions?**



# **Transport properties**

Modeling the effect of temperature on diffusion coefficients:

$$D_i = D_i^{ref} \exp \left[ \alpha \left( T - T^{ref} \right) \right]$$

#### Evaluation of $\underline{\alpha}$ :

W/C	Hydration (days)	α(1)	
		Type 10	Type 50
0.45	28	0.0248	0.0173
	91	0.0207	0.0272
	365	0.0314	0.0377
0.65	28	0.0283	0.0277
	91	0.0372	0.0346
	365	0.0341	0.0446
0.75	28	0.0286	0.0299
	91	0.0320	0.0280
	365	0.0313	0.0309



## **Transport properties**

Modeling the effect of temperature on diffusion coefficients:

$$D_i = D_i^{ref} \exp \left[ \alpha \left( T - T^{ref} \right) \right]$$

The value of  $\underline{\alpha}$ :

- does not depend on the w/c,
- does not depend on the type of cement,
- does not depend on hydration.

The parameter  $\underline{\alpha}$  characterizes the effect of temperature on diffusion.

The global analysis of the results gives:  $\alpha = 0.028$ .

### **Heat transfer**

The following heat conduction equation is implemented in the 1D version of the model:

$$\rho C \frac{\partial T}{\partial t} - div \left( k \operatorname{grad} T \right) = 0$$

where the conductivity  $\underline{k}$  depends on temperature  $\underline{I}$  and the degree of saturation  $\underline{S}^*$ :

$$k = k^{ref} (0.244(S-1)+1) \times (0.0015(T-T^{ref})+1)$$



# **Transport equations**

Without the constant temperature assumption, the following term is added to the flux relationship:

$$j_{i} = -D_{i} \frac{\partial c_{i}}{\partial x} - \frac{D_{i} z_{i} F}{RT} c_{i} \frac{\partial \psi}{\partial x} - D_{i} c_{i} \frac{\partial \ln \gamma_{i}}{\partial x} \left( -\frac{D_{i} c_{i} \ln(\gamma_{i} c_{i})}{T} \frac{\partial T}{\partial x} \right)$$

•Special care must be taken when  $c_i=0$  because of the ln term. Evaluating the limit shows that the term tends to 0 in that case.



### Chemical reactions

The effect of temperature on chemical reactions is modeled according to the **Van't Hoff relationship**.

It relates the equilibrium constant of the solid phases considered in **STADIUM**<sup>®</sup> with temperature.

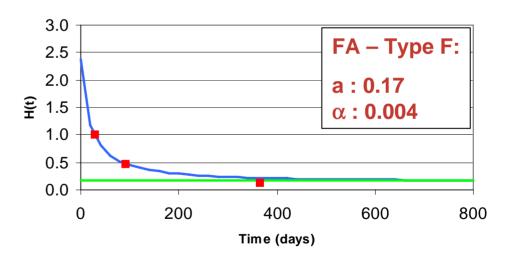
$$\ln(K_{\rm sp}) = \ln(K^o) + \frac{\Delta H^o}{R} \left( \frac{1}{T^o} - \frac{1}{T} \right)$$

- To and Ko are reference values at 25°C.
- △H° is the reaction enthalpy.



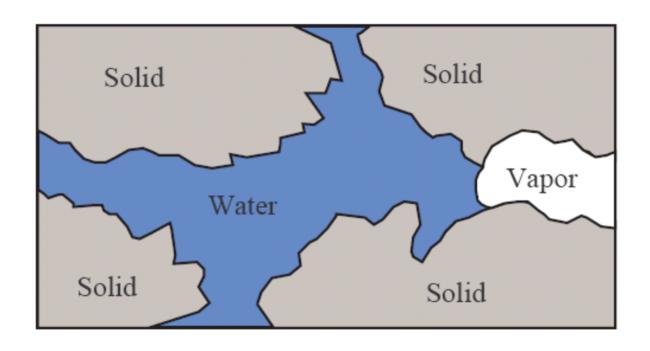
Hydration function: 
$$H(t) = \frac{a}{1 + (a-1)e^{-\alpha(t-t_{ref})}}$$

	D <sub>Cl</sub> (E x 10 <sup>-11</sup> m <sup>2</sup> /s)
Curing	Type 10
28	15,2
91	14,9
365	14,1





### Impact of moisture content

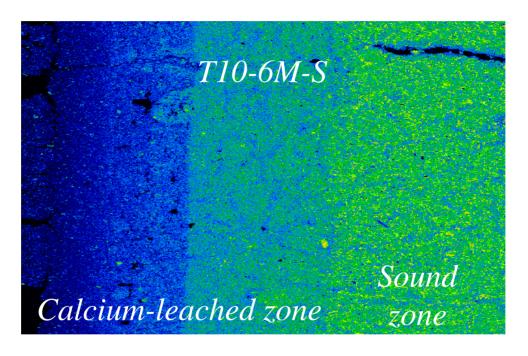


$$D_i = \tau D_i^o \left(\frac{w^{7/3}}{\phi^{7/3}}\right)$$

Millington and Quirk, Trans. Faraday Soc., vol. 57 (1961)



## Impact of degradation



$$H_D(\phi) = \left(\frac{\phi}{\phi_o}\right)^3 \left(\frac{1-\phi_o}{1-\phi}\right)^2 \qquad H_D(\phi) = \frac{e^{4.3\phi/V_p}}{e^{4.3\phi_o/V_p}}$$

